

METRO
MEtallurgical TRaining On-line



Basic phenomena accompanying alloy solidification

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Education and Culture



Surface Energy



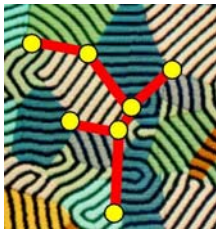
Definition of surface energy

$$\gamma = \left(\frac{\partial G}{\partial A} \right)_{T,p}$$

Where:

G - Gibbs free energy

A - Surface area

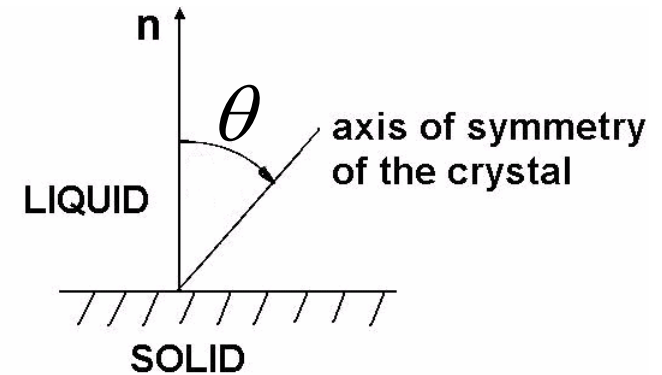


Surface Energy



Dependence of surface energy on different factors

$$\gamma = \gamma(T, C, \theta)$$



where:

T - temperature

C - concentration of species

θ - angle between normal to the surface and crystallographic axis



Surface Energy



Dependence of surface energy on angle from normal to the interface

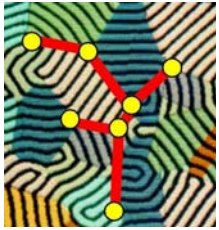
$$\gamma(\theta, T, C) = \gamma_0(T, C) \left[1 + A_s \left[\frac{8}{3} \sin^4 \left(\frac{1}{2} m_s (\theta - \phi_s) \right) - 1 \right] \right]$$

Where:

A_s - magnitude of anisotropy

ϕ_s - angle of symmetry axis with respect to the interface normal

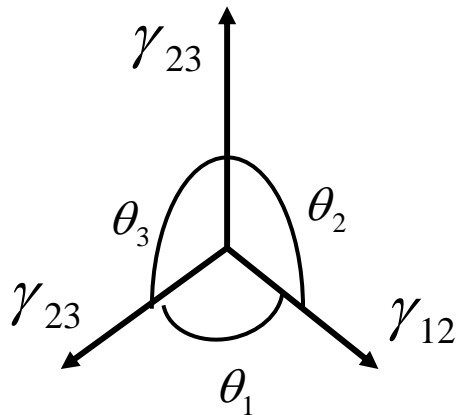
m_s - mode of symmetry of the crystal



Surface Energy



Relative surface energies between three phases



$$\frac{\sin \gamma_{23}}{\theta_1} = \frac{\sin \gamma_{13}}{\theta_2} = \frac{\sin \gamma_{12}}{\theta_3}$$

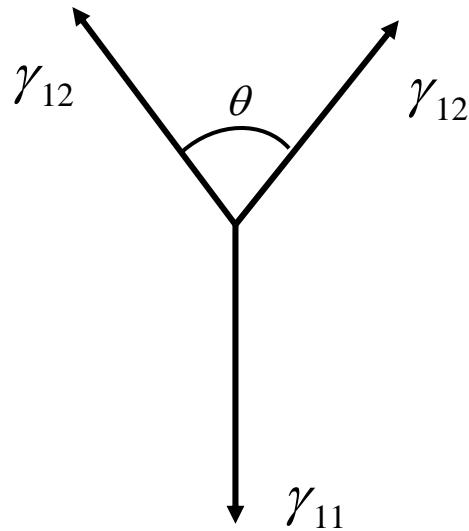


Surface Energy

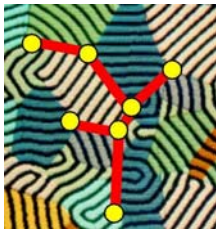


Measurement of surface energy in solids

Thermal etching or grooving



$$\gamma_{11} = 2\gamma_{12} \cos(\theta / 2)$$

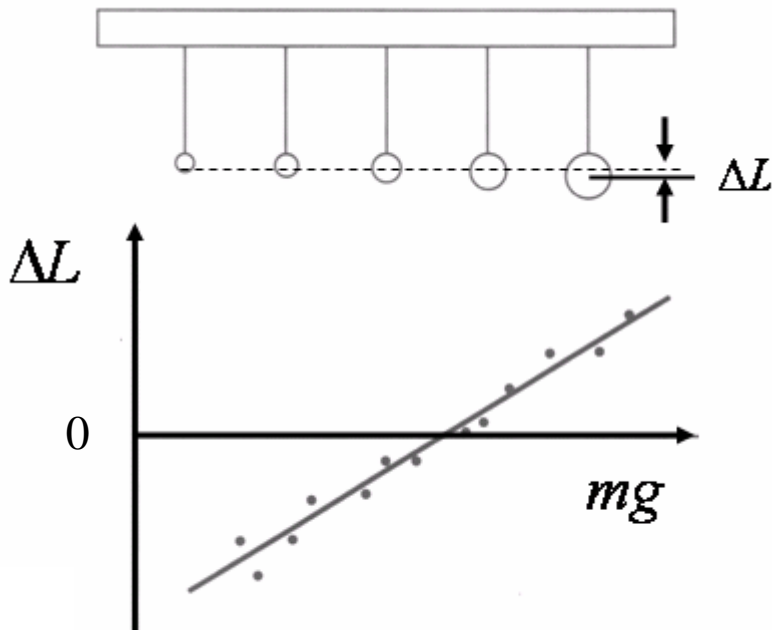


Surface Energy



Measurement of surface energy in solids

Elongation method

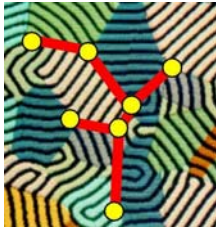


$$\gamma_{sg} = \frac{mg}{\pi r}$$

where:

mg - weight applied

r - wire radius



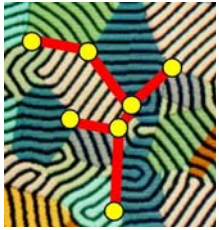
Surface Energy



Effect of temperature on the surface energy

$$\left(\frac{\partial \gamma}{\partial T} \right)_{A,p} = S_A$$

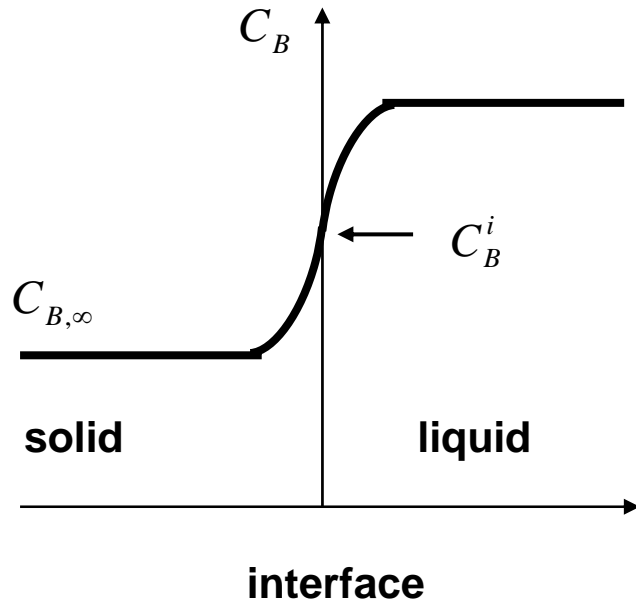
where: S_A - surface entropy



Surface Energy



Segregation of components at the interface



$$\Gamma_B^i = -\frac{1}{RT} \left(\frac{\partial \gamma}{\partial x_B} \right)_T$$

where:

Γ_B^i - surface concentration, $\Gamma_B^i = C_B^i - C_{B,\infty}$

x_B - mole fraction of the solute

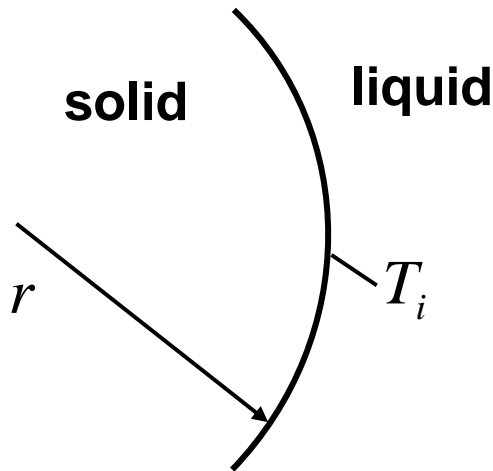
R - universal gas constant



Surface Energy



Solidification temperature of the curved interface



$$T_i = T_m - \frac{2V_{ms}\gamma_{ls}}{\Delta S_m} \kappa$$

where:

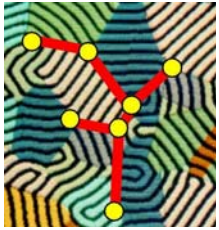
T_i - interface temperature

T_m - equilibrium solidification temperature

V_{ms} - molar volume of the solid phase

ΔS_m - entropy change during solidification per mole

κ - interface curvature, $\kappa = r^{-1}$



Surface Energy



Solidification temperature of the curved interface

$$\Delta S_m = \frac{\Delta H_m}{T_m} = \frac{L_m}{T_m}$$

where:

ΔH_m - enthalpy change during solidification per mole

L_m - latent heat of solidification per mole



Surface Energy



Solidification temperature of the curved interface

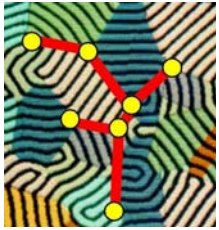
$$T_i = T_m + m_l C_l - \frac{2V_{ms}\gamma_{ls}}{\Delta S_m} \kappa$$

where:

m_l - liquidus slope

C_l - solute concentration in the liquid phase

κ - interface curvature, $\kappa = \frac{1}{2} \left(\frac{1}{r_1} + \frac{1}{r_2} \right)$

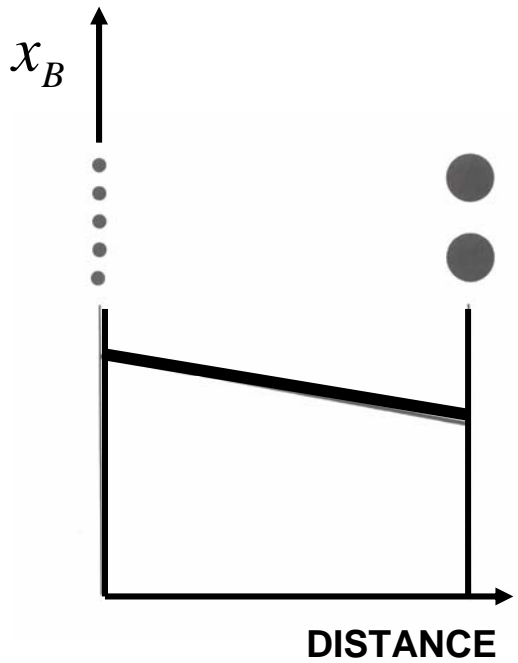


Surface Energy



Coarsening of solid grains

$$x_B(r) = x_B(r \rightarrow \infty) \exp\left[\frac{2V_{ms}\gamma_{\alpha\beta}}{RT} \kappa\right]$$

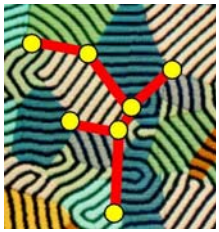


where:

$x_B(r)$ - mole fraction of component B in the solid phase for curved interface

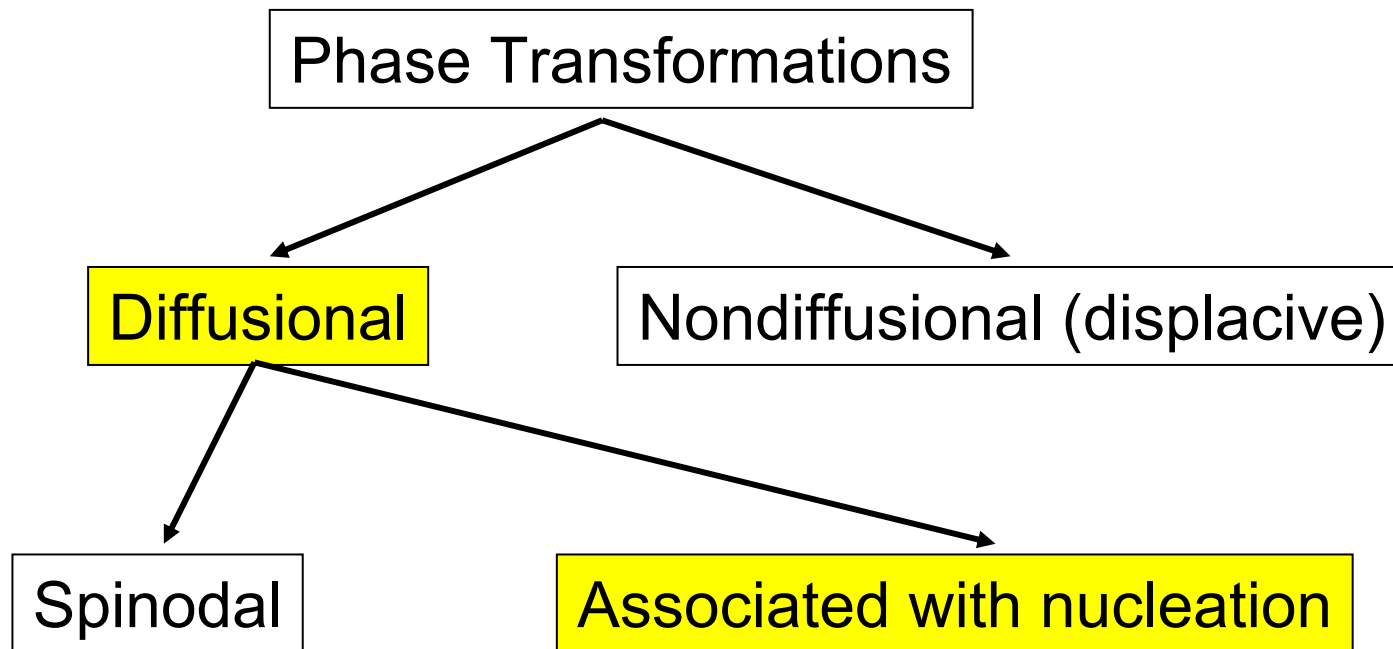
$x_B(r \rightarrow \infty)$ - mole fraction of component B in the solid phase for plane interface

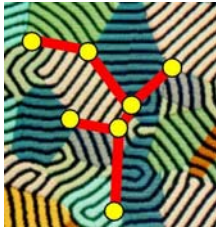
$\gamma_{\alpha\beta}$ - surface energy between phases α and β



Nucleation of solid phase

Types of phase change transformations

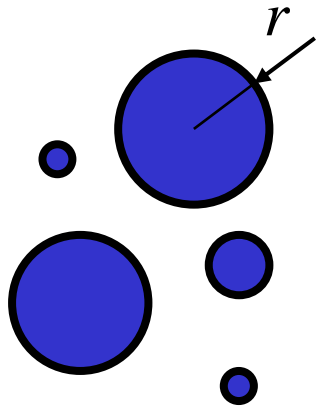




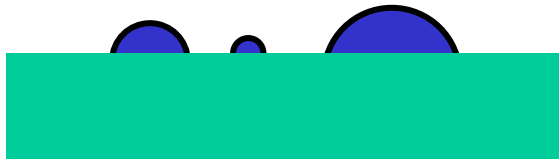
Nucleation of solid phase



Types of nucleation



- homogeneous



- heterogeneous



Nucleation of solid phase



Size distribution of nuclei

$$n(r) = n_o \exp\left(-\frac{\Delta G_f}{kT}\right)$$

where:

$n(r)$ - number of particles clusters of radius r *per unit volume*

n_o - number of liquid particles *per unit volume*

ΔG_f - Gibbs free energy of formation of the new phase

k - Boltzmann constant, $k=1.38 \cdot 10^{-23}$ J/K



Nucleation of solid phase



Size distribution of nuclei for solidification temperature

$$\Delta G_f = 4\pi r^2 \gamma_{sl}$$

hence

$$n(r) = n_o \exp\left(-\frac{4\pi r^2 \gamma_{sl}}{kT_m}\right)$$



Nucleation of solid phase

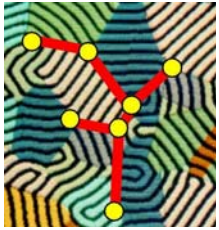


Gibbs free energy change associated with phase transformation

$$\Delta G_r = \frac{4\pi}{3} r^3 \Delta G_v + 4\pi r^2 \gamma_{sl}$$

where:

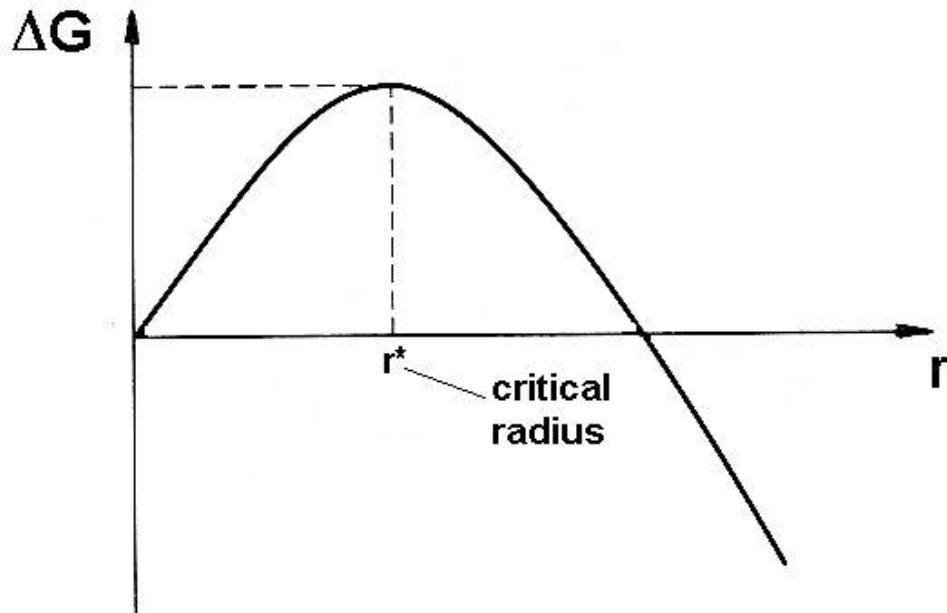
ΔG_v - number of particles clusters of radius r *per unit volume*



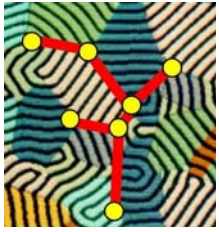
Nucleation of solid phase



Minimum radius of nuclei

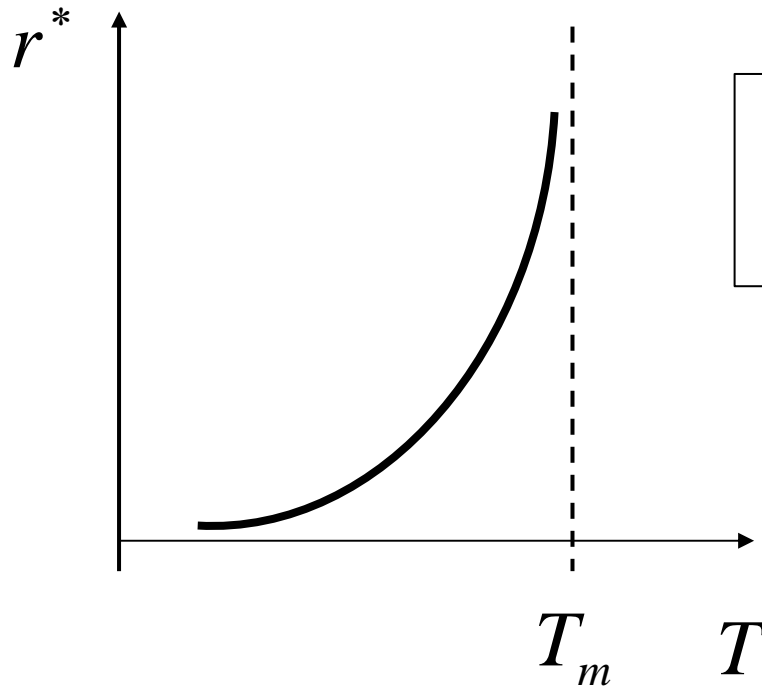


$$r^* = -\frac{2\gamma_{sl}}{\Delta G_v}$$



Nucleation of solid phase

Influence of undercooling on critical radius



$$r^* = -\frac{2\gamma_{sl}T_m V_m}{L_m(T - T_m)}$$

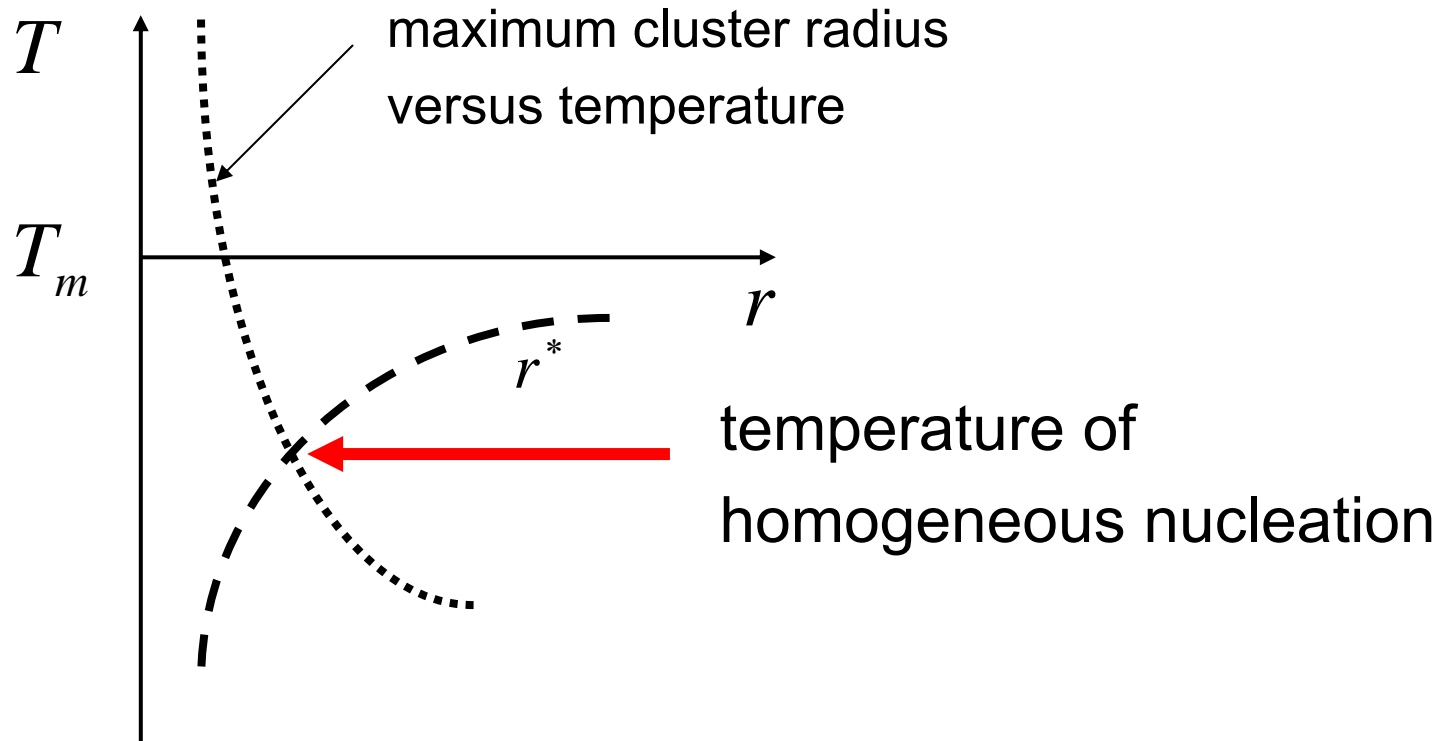
where:

V_m - molar volume



Nucleation of solid phase

Influence of undercooling on critical radius

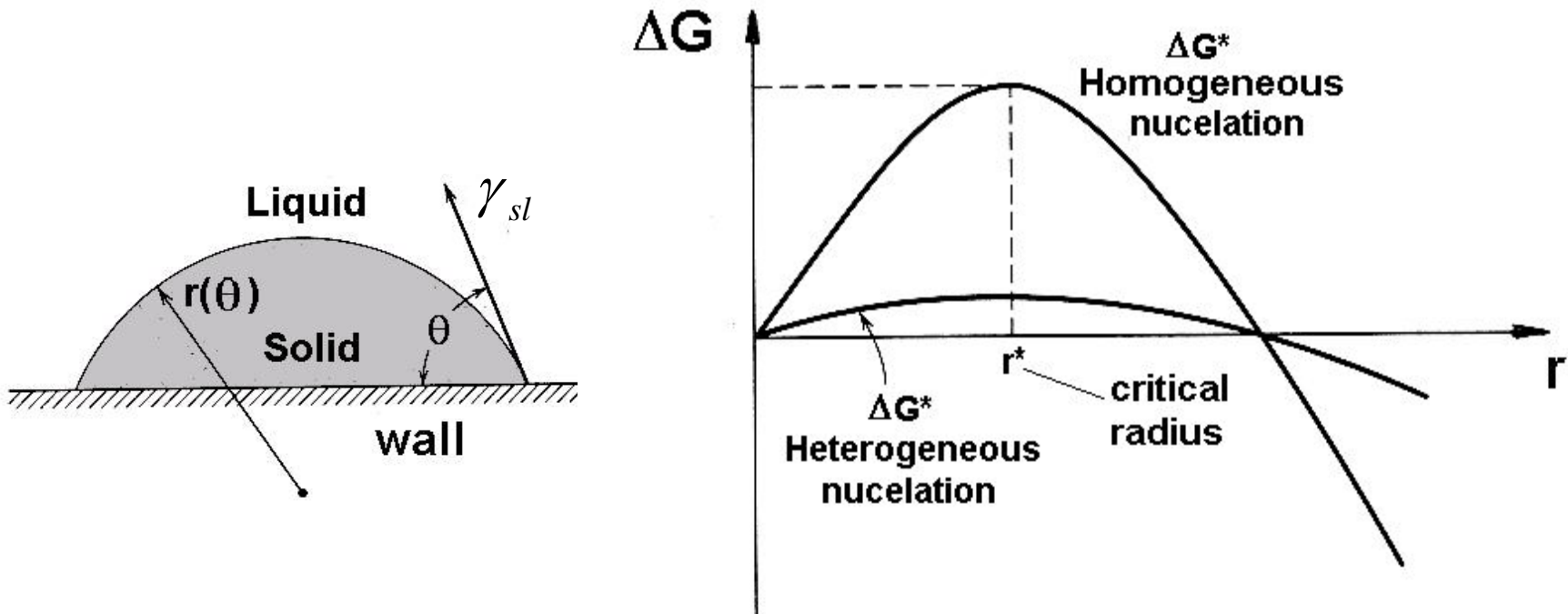


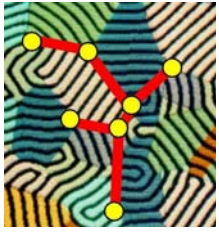


Nucleation of solid phase



Influence of mould surface on nucleation





Nonequilibrium solidification



Velocity of the liquid/solid interface

$$w_i(T_i, C_s, C_l) = w_c(T_i) \left[1 - \exp(\Delta G_m(T_i, C_s, C_l) / RT) \right]$$

where: ΔG_m - Gibbs free energy change per mole of alloy solidified

w_c - characteristic speed of crystallization

C_s - composition of the growing solid

C_l - composition of the liquid at the interface

T_i - interface temperature



Nonequilibrium solidification



Partition coefficient for nonequilibrium solidification

$$K_p(w_i) = \frac{K_{pe} + w_i / w_D}{1 + w_i / w_D}$$

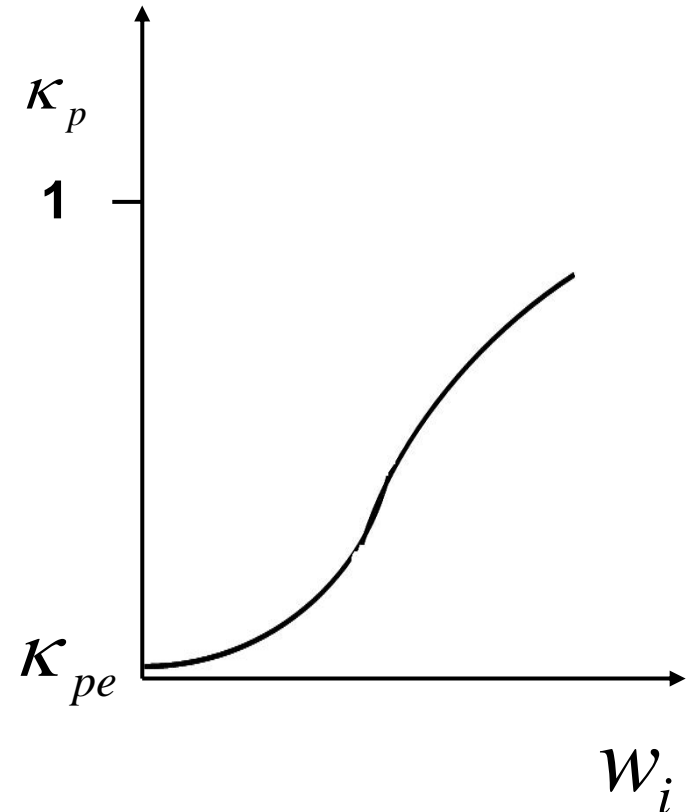
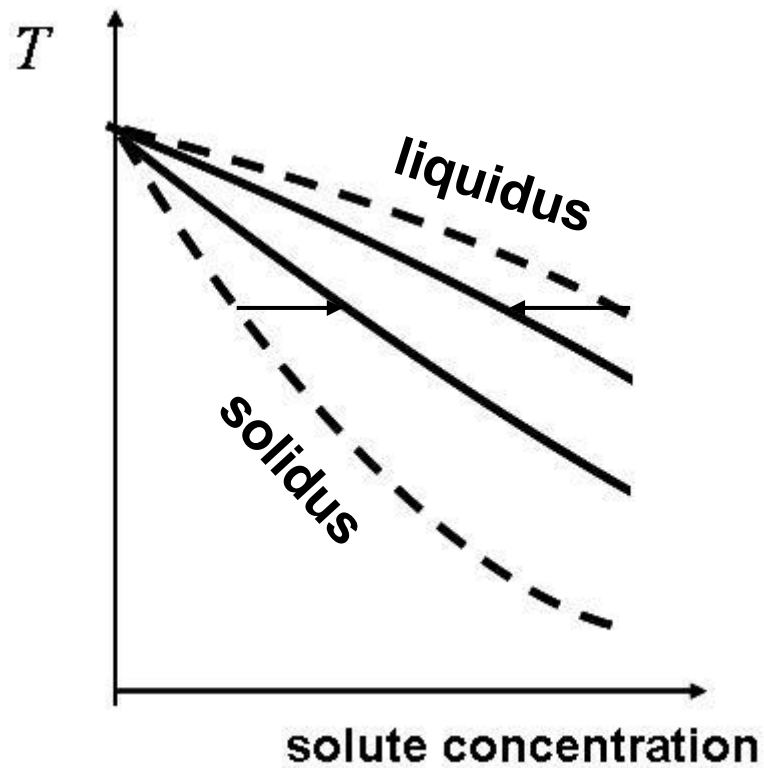
- where:
- K_p - partition coefficient for non-equilibrium solidification
 - K_{pe} - partition coefficient for equilibrium solidification
 - w_i - velocity of liquid/solid interface
 - w_D - diffusive speed



Nonequilibrium solidification



Partition coefficient for nonequilibrium solidification



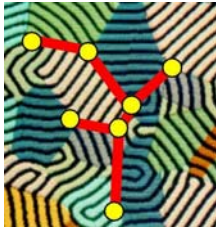


Nonequilibrium solidification

Interface temperature during rapid solidification

$$T_i = T_m + m_l (w_i) C_l - \frac{RT^2}{L_m} \frac{w_i}{w_s}$$

- where:
- R - universal gas constant
 - L_m - latent heat of solidification
 - w_i - velocity of liquid/solid interface
 - w_s - sound velocity
 - C_l - solute concentration
 - m_l - slope of the liquidus line



Nonequilibrium solidification



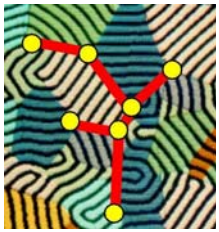
Slope of the liquidus line during rapid solidification

$$m_l(w_i) = m_l^e \frac{1 - \kappa_p(w_i) [1 - \ln(\kappa_p(w_i) / k_e)]}{1 - \kappa_{pe}}$$

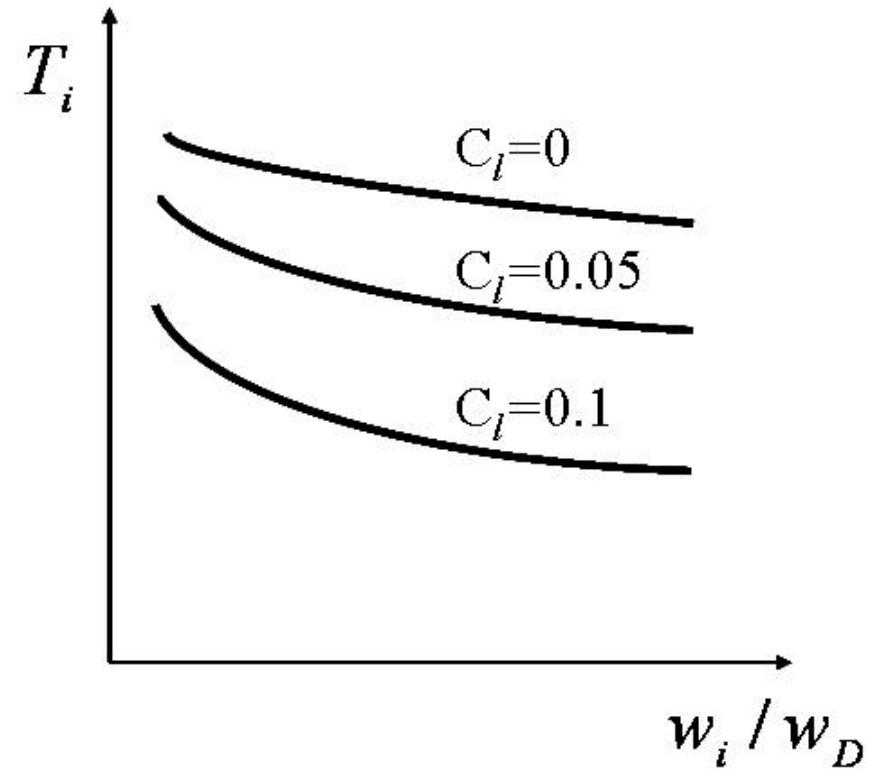
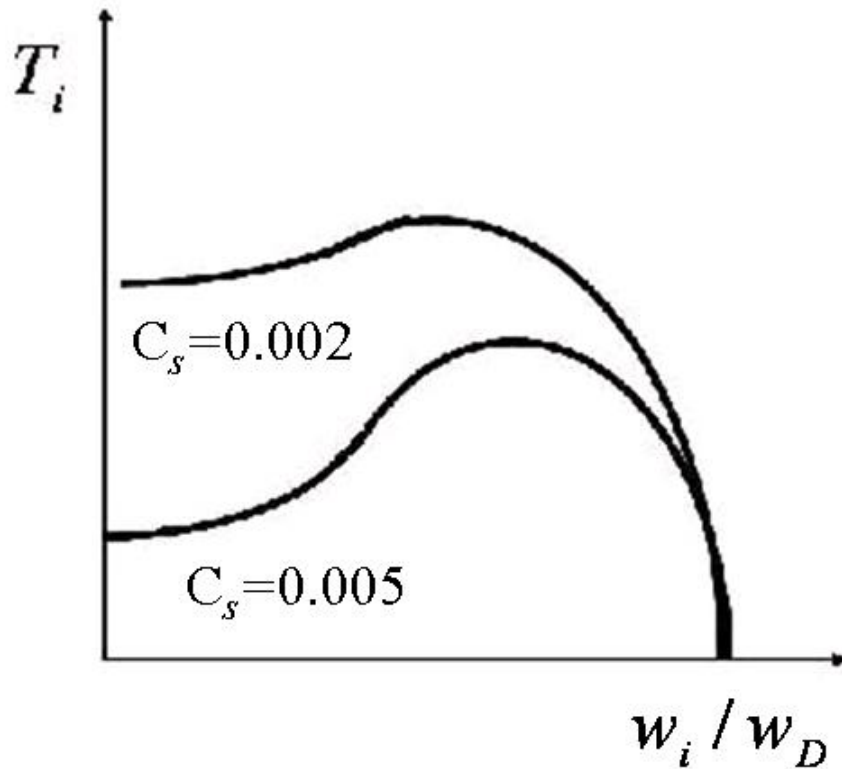
where: m_l^e - slope of the liquidus line in the case of equilibrium

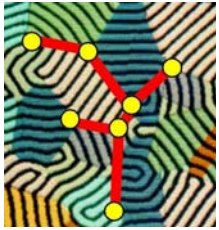
κ_{pe} - equilibrium partition factor

$\kappa_p(w_i)$ - nonequilibrium (kinetic) partition factor



Nonequilibrium solidification





Nonequilibrium solidification



Interface temperature during rapid solidification for curved interface

$$T_i = T_m + m_l(w_i)C_l - \frac{RT^2}{L_m} \frac{w_i}{w_s} - \frac{2\gamma_{ls}V_aT_m}{L_m} \kappa$$

where: V_a - atomic volume

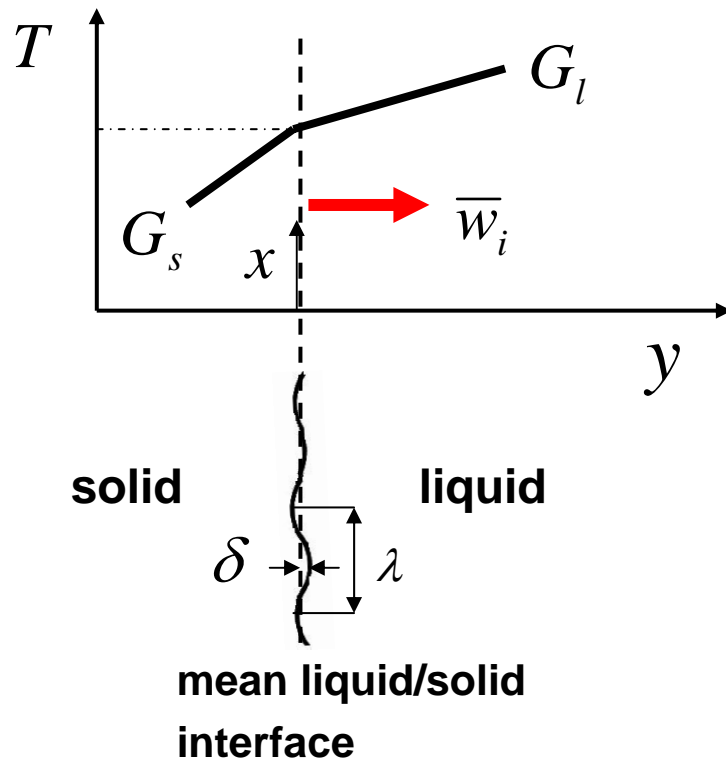
κ - interface curvature



Instability of alloy solidification



Instability of the planar liquid/solid interface



where:

G_s - temperature gradient in the solid

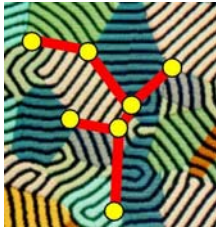
G_l - temperature gradient in the liquid

w_i - interface velocity

Sinusoidal perturbation
of the solid/liquid interface

$$y = \delta \sin \omega x$$

where: $\omega = \frac{2\pi}{\lambda}$



Instability of alloy solidification



Stability criterion

$$\frac{\left(\frac{\partial \delta}{\partial t}\right)}{\delta} < 0$$

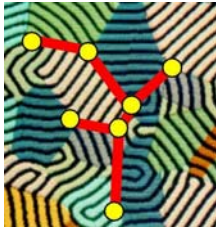


stable interface

$$\frac{\left(\frac{\partial \delta}{\partial t}\right)}{\delta} > 0$$



unstable interface



Instability of alloy solidification



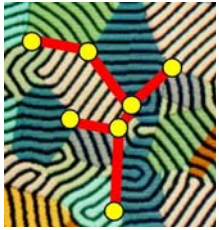
Influence of different factors on stability

For negligible G_l

$$\frac{\left(\frac{\partial \delta}{\partial t}\right)}{\delta} = f(V, Q, \tilde{\omega})$$

where:

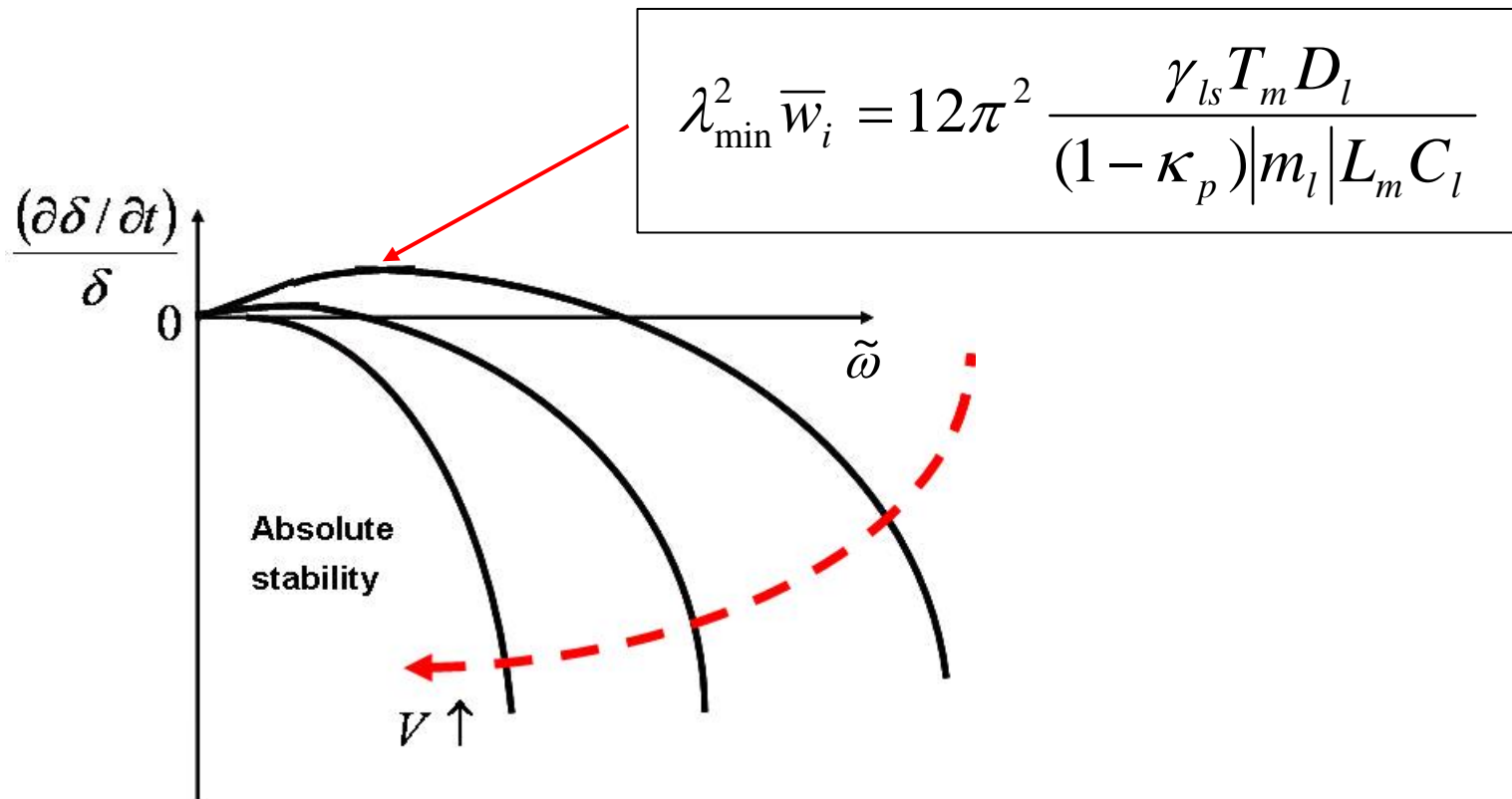
$$V = \frac{\gamma_{sl} T_m k_s \bar{w}_i}{D_l L_m^2}, \quad Q = \frac{\kappa_p D_l L_m}{m_l C_l k_s}, \quad \tilde{\omega} = \omega \frac{D_l}{\bar{w}_i}$$



Instability of alloy solidification



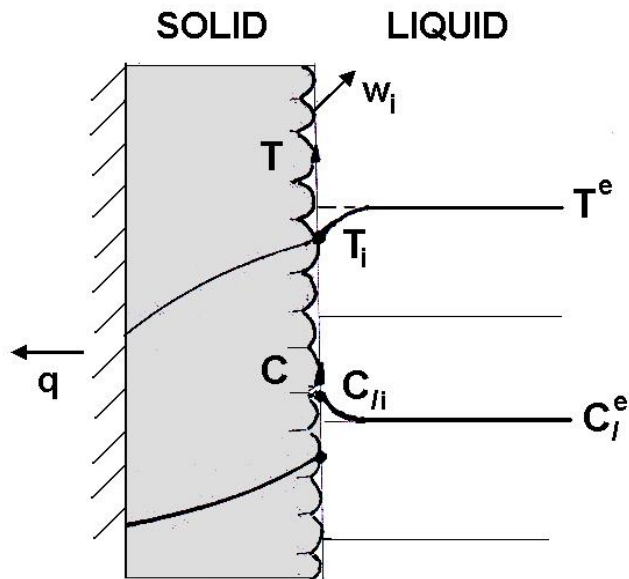
Influence of different factors on stability



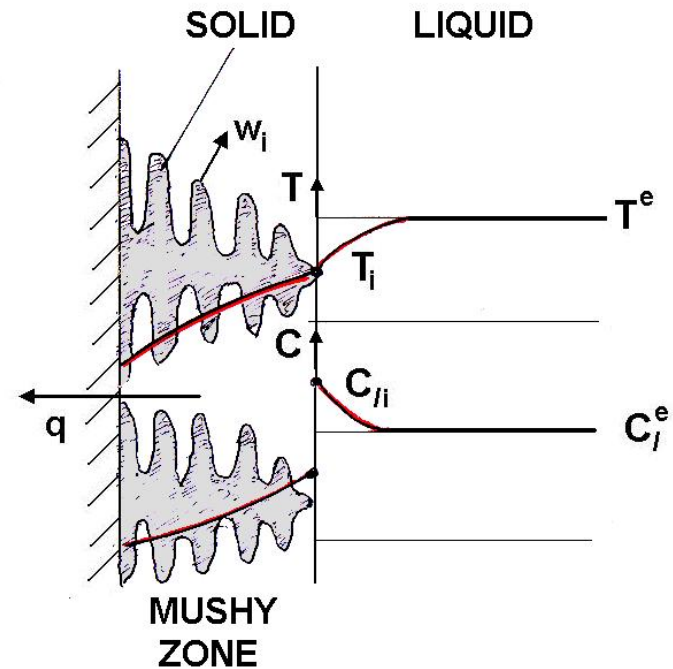


Formation of mushy zone

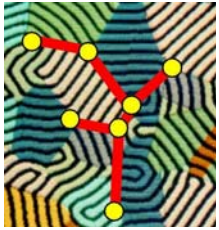
Growth of dendritic structure



columnar growth



columnar dendritic growth



Formation of mushy zone

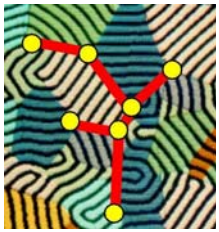


Critical thermal gradient in the liquid at which the cellular microstructure pattern can be developed

$$\left(\frac{\partial T}{\partial n} \right)_l \geq \frac{m_l C_0 (1 - \kappa_p)}{\kappa_p D_l} w_i$$

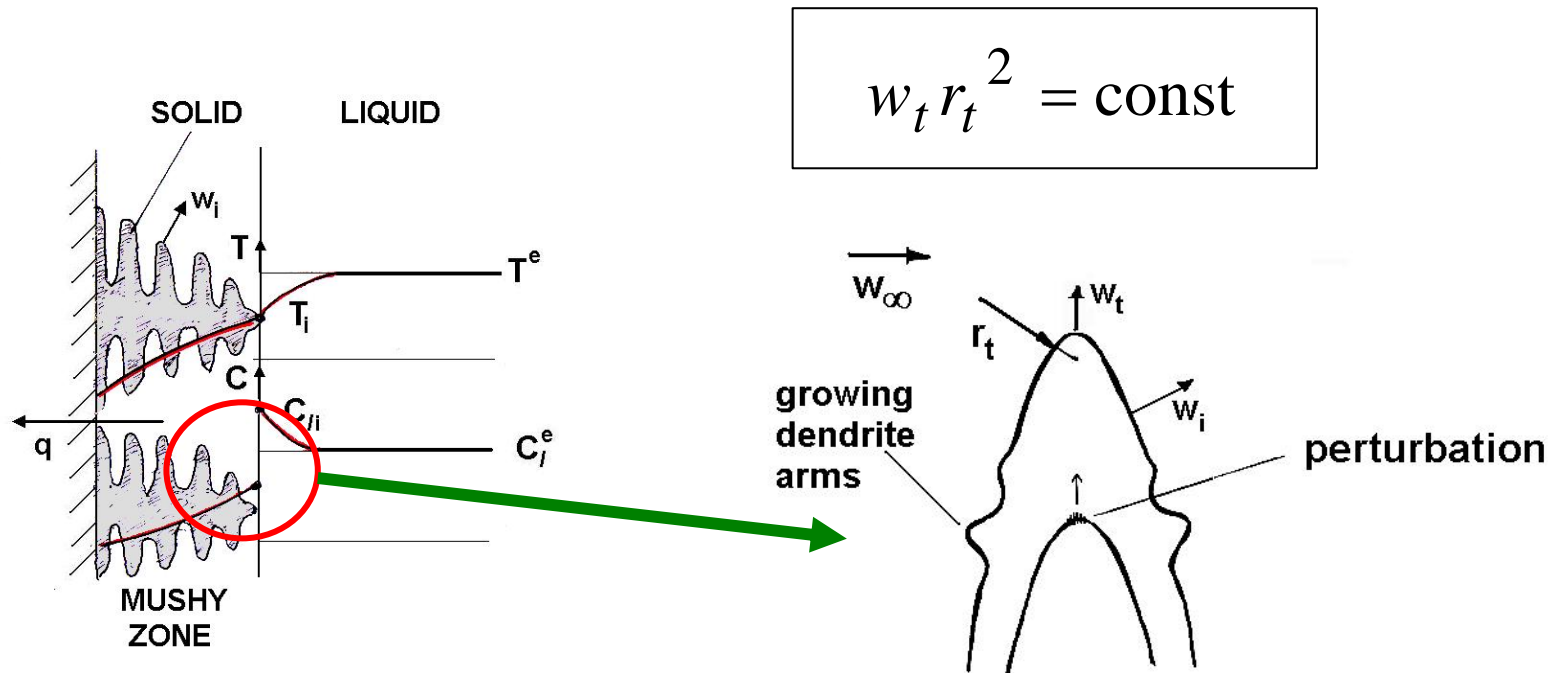
where:

C_0 - initial solute concentration



Formation of mushy zone

Growth a separate columnar dendrite





Formation of mushy zone

Undercooling of a dendrite with a parabolic tip

$$\Delta T_t = \frac{L_m I\nu(\text{Pe})}{c_{pl}} + m_l \left[1 - \frac{1}{1 - (1 - \kappa_p) I\nu(\text{Pe})} \right] C_0 + 2 \frac{\Gamma}{r_t}$$

where $Pe_t = \frac{w_t r_t}{a_l}$ - Peclet number, Γ - Gibbs-Thomson coefficient,

c_{pl} - liquid specific heat, a_l - liquid thermal diffusivity

$I\nu(Pe)$ - Ivanstov function, $I\nu(Pe) = Pe \exp(Pe) E(Pe)$



Formation of mushy zone

Tip radius of a dendrite with a parabolic tip

$$r_t = \frac{\frac{\Gamma}{\sigma^*}}{\frac{Pe L_m}{c_{pl}} + \frac{Sc m_l C_0 (1 - \kappa_p)}{[1 - (1 - \kappa_p) Iv(Sc)]}}$$

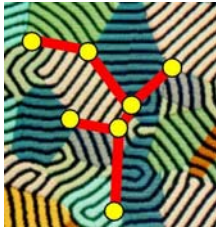
where

Sc - Schmit number

$$Sc = \frac{w_t r_t}{D_l}$$

D_l - solute diffusion coefficient in the liquid,

σ^* - constant following from the marginal theory



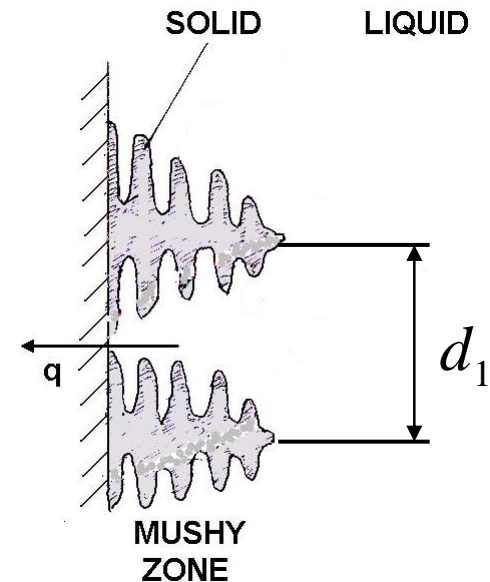
Formation of mushy zone

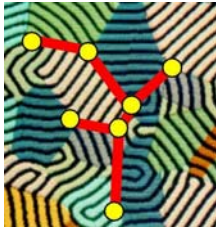
The primary arms spacing

$$d_1 = \frac{A(D_l \Gamma)^{1/4}}{\sqrt{\left(\frac{\partial T}{\partial n}\right)_l w_t r_t}}$$

where

A - a numerical constant.

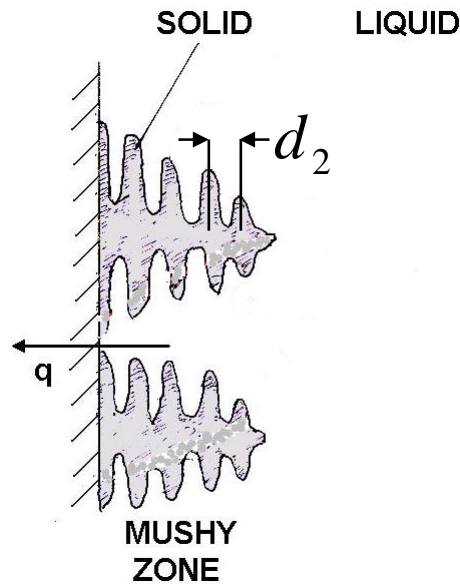




Formation of mushy zone



The secondary arms spacing



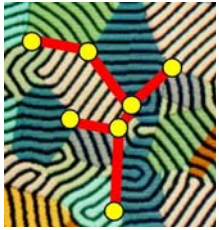
$$d_2 = \sqrt[3]{l_T l_C d_o}$$

where

l_T - thermal field lengthscale

l_C - solutal field lengthscale

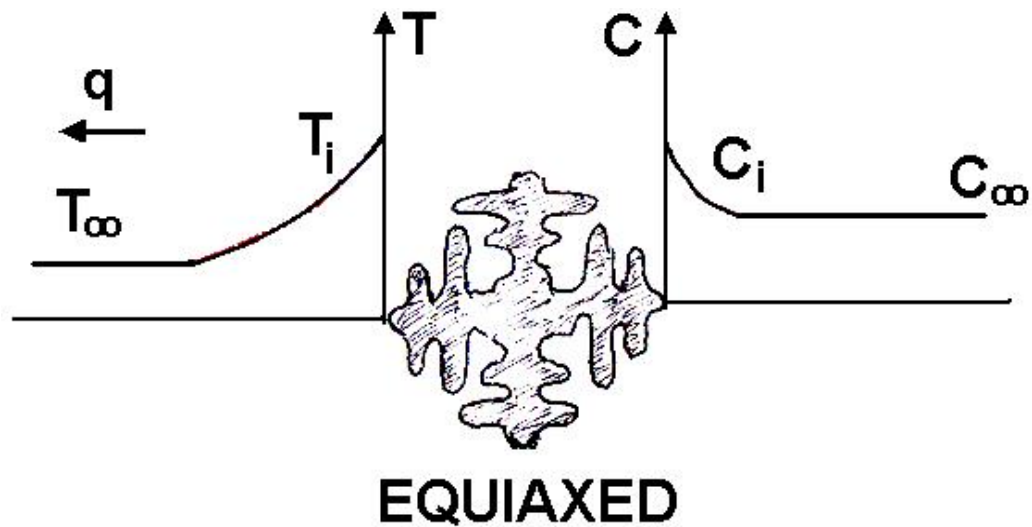
d_o - capillary lengthscale

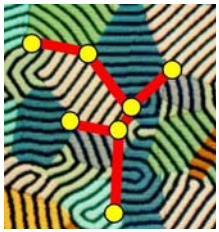


Nonequilibrium solidification



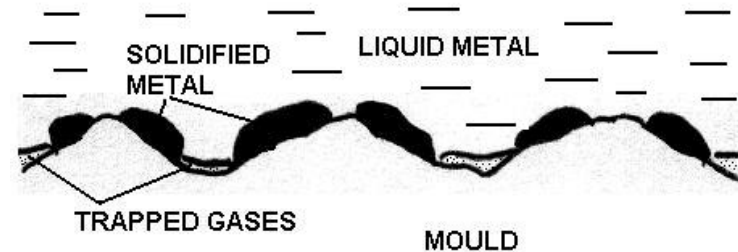
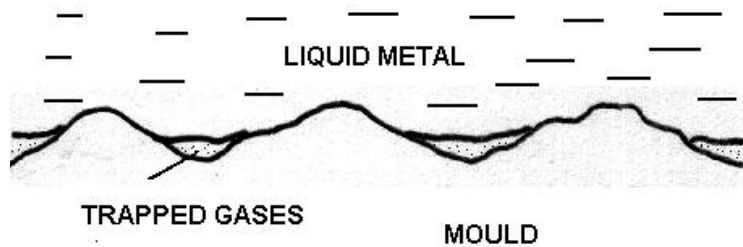
Growth of separate equiaxed grains



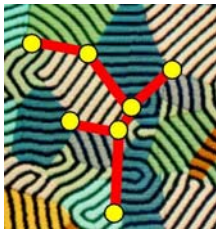


Mould-melt interaction

Growth of solid on the mould surface



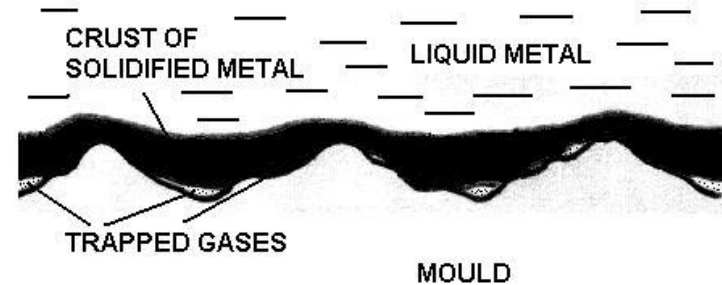
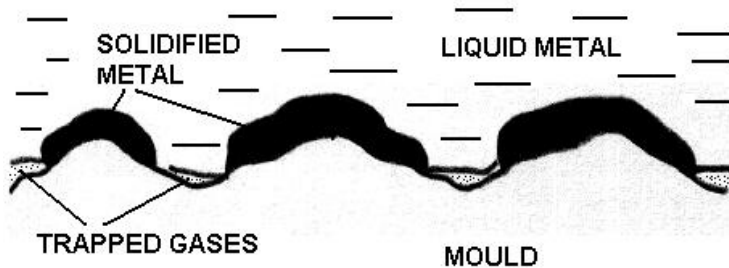
Evolution of the solidification process at the alloy-mould interface



Mould-melt interaction



Growth of solid on the mould surface



Evolution of the solidification process at the alloy-mould interface



Summary



Basic phenomena associated with solidification of pure metals and alloys

- surface energy
- nucleation of solid phase
- non-equilibrium effects during solidification
- instability of the solid/liquid interface
- formation of the mushy zone
- mould/melt interactions